

## 6. Observing Dislocations and Other Defects

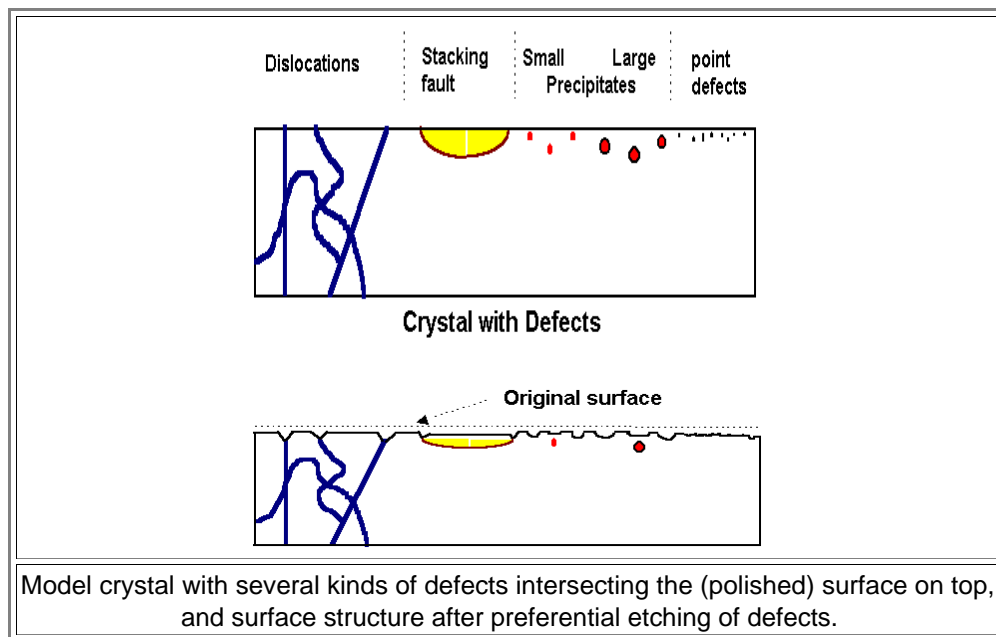
### 6.1 Decoration and Conventional Microscopy

#### 6.1.1 Preferential Etching

##### Basics of Preferential Etching

The basic idea behind **preferential etching** is to mark defects intersecting the surface by a small pit or groove, so they become visible in a microscope.

- Start with a well polished surface that does not show any structures in a **light microscope** (including high magnifications and sensitive modes, e.g. phase or interference contrast)
- Find an etching solution that dissolves your material much more quickly around defects than in perfect regions (that is the tricky part).
- Expose (= etch) your sample in this solution for an appropriate amount of time. What happens will be something like this:



After preferential etching you obtain well developed **etch pits** (actually something looking more like pointed etch cones) at the intersection points of dislocations (including partial dislocations) and the surface and **etch grooves** at the intersection line of grain boundaries and stacking faults with the surface. Precipitates will be shown as shallow pits with varying size, depending on the size of the precipitate and its location in the removed surface layer. Areas with high densities of very small precipitates may just appear rough. Two-dimensional defects as grain boundaries and stacking faults may be delineated as grooves.

- There is a certain problem with grain boundaries, however: They may also be *delineated*, i.e. rendered visible, with chemicals that do not preferentially etch defects, but simply dissolve the material with a dissolution velocity that depends on the grain orientation (this is the rule and not the exception for most chemicals).
- In this case grain boundaries show up as *steps* and not as *grooves*. Small steps and grooves, however, look very similar in a light microscope and may easily be mixed up.

You may think: So what! - in any case I see the grain boundary. Well, almost right, but not quite - there are problems:

- Grain boundaries separating two grains with similar orientation with respect to the surface would not be revealed.
- The delineation of grain boundaries obtained under uncertain etching conditions suggests that you delineated *all defects* - but in fact you did not. Delineation of grain boundaries thus must not be taken as an indication that the etching procedure works and there are no defects, because you don't see any!

Before we look at examples and case studies, two important points must be made:

1. Defect etching for many scientists is a paradigm for "**black art**" in science. There are good reasons for this view:

- Nobody knows how to mix a preferential etching solution for some material from theoretical concepts. Of course you must look for chemicals or mixtures of chemicals that react with your material, but not too strongly. But after this bit of scientific advice you are on your own in trying to find a suitable preferential etch for your material.

- Well-established preferential etching solutions usually have unknown and poorly understood properties. They sometimes work only on specific crystallographic orientations; their detection limits for small precipitates are usually unknown; they may also depend on other parameters like the doping level in semiconductors; and so on.

## 2. Defect etching *in practice* is more *art* than science.

- Beginners, even under close supervision by a master of the art, will invariably produce etched samples with rich structures that have nothing to do with defects - they produced so-called **etch artifacts**. It takes some practice to produce reliable results.
- But:** Defect etching still is by far the most important and often most sensitive technique for observing and detecting defects!

There are many routine procedures for delineating the defects structure of metals by etching. Here we will focus on defects etching in Silicon; which is still the major technique for defect investigations in **Si** technology. Some [details and peculiarities of defect etching in Si](#) can be found in the link. In what follows we look at the power and possible mechanisms of preferential etching in the context of examples from recent research.

### Defect Etching Applied to Swirl Defects in Silicon:

The name "Swirl defects" was used for grown-in defects in large **Si** crystals obtained by the float-zone technique in the seventies.

- Swirl defects are a subspecies of what now is known as "**bulk micro defects**" (**BMD**); they are nothing but agglomerates of the point defects present in thermal equilibrium near the melting point with possible influences of supersaturated impurities still present in ultra clean **Si** (only oxygen and on occasion carbon).
- Whereas the relatively large swirl defects are no longer present in state-of-the-art **Si** crystals, point defect agglomerates and oxygen precipitates still are - there is no way to eliminate the equilibrium defects! **BMDs** are a major concern in the **Si** industry because they cause malfunctions of integrated circuits. The link leads to some [recent papers on point defects and BMDs in Si crystals](#).

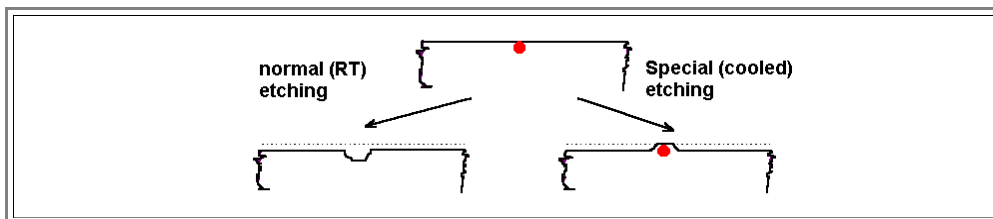
Most of the examples relating to **Si** are taken from the work of **B.O. Kolbesen** (formerly at Siemens; now (2001) at the University of Frankfurt).

- The name "swirl" comes from the spiral "swirl-like" pattern observed in many cases by *preferential etching* as shown on the right.
- Close inspection revealed two types of etch features which must have been caused by different kinds of defects. Lacking any information about the precise nature of the defects (which etching can not give), they were termed "**A-**" and "**B-swirl** defects". More [pictures and information](#) in the link



Understanding the precise nature of swirl defects was deemed to be very important for developing crystal growth techniques that could avoid these detrimental defects.

- But etching alone can not give structural data, and other techniques as, e.g., transmission electron microscopy, could not be applied directly because the densities of swirl defects was too small (the likelihood of having a defect in a typical **TEM** sample was practically zero). A combination of a special etching technique and **TEM**, however, could give the desired results.
- The power and the "black art" component of defect etching is nicely demonstrated by the following development: A "special etch" which was simply the old solution, but cooled to about freezing temperatures, did not produce etch pits (and thus remove the defect) for A-swirls, but hillocks (still containing the defect).

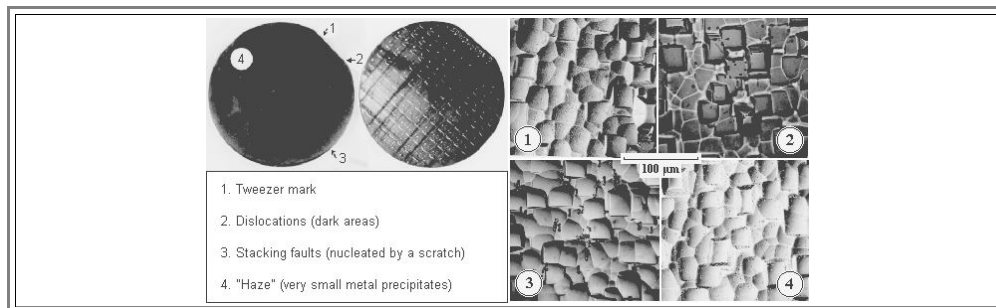


The hillocks identified the precise location of the **A-swirl** defect. A special preparation technique rendered the areas containing hillocks transparent for **TEM** investigations, and the structure of **A-swirls** defects could be identified. They consisted of dislocation loop arrangements that were generated by the agglomeration of interstitials. This gave the first direct evidence that self-interstitials are important in **Si**.

- B-swirl** defects could not be identified with this technique - their nature is still not clear.
- [More about swirl defects](#) and the application of preferential etching can be found in an original paper (in German) in the link.

## Process Control by Etching Defects during the Manufacture of Integrated Circuits

- ▶ The manufacture of **integrated circuits (IC)** involves many processes prone to introduce defects in the more or less perfect starting crystal.
  - All high temperature processes induce temperature gradients which lead to stress and thus to a driving force for plastic deformation. Since the starting material is dislocation free, the decisive process is the generation of the first dislocations which is much easier if small precipitates or dislocation lops are already present.
  - Thermal oxidation introduces **Si** interstitials with a strong tendency to agglomerate into stacking fault loops, so-called **oxidation induced stacking faults (OSF)**.
  - All processes tend to induce trace amount of metals which will diffuse into the **Si** and eventually precipitate.
  - Ion implantation destroys the lattice to a large degree up to complete amorphization. Even upon careful annealing some defects may be left over.
- ▶ As a general rule, all defects in the electronically active part of an **IC** (roughly the the first **5 μm - 10 μm** of the wafer) are deadly for the device. They have to be avoided and that means that they have to be monitored first. The method of choice is **preferential etching**.
- ▶ Lets look at an example
  - The pictures show a **Si** wafer with several defect types introduced during very early stages of processing. [Details](#) are provided in the link.

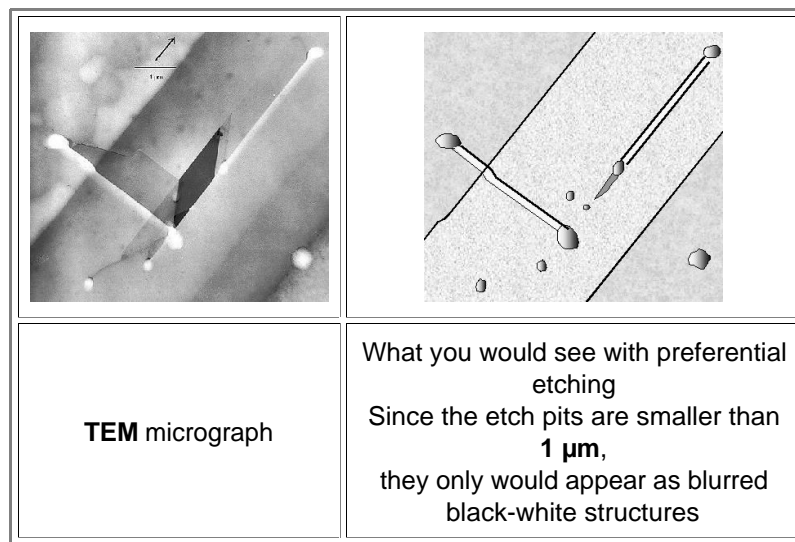


- ▶ A few more example are provided in the links. They might be a bit unconvincing, but be aware that looking into an actual microscope gives you much more information than what can be captured in a few pictures.
  - [Development of stacking faults in bipolar transistors](#)
  - [Precipitates and other defects](#)

▶ We are now able to compare weaknesses and strength of preferential etching for defect detection:

| Strength   | Weaknesses   |
|--|--|
| <ul style="list-style-type: none"> <li>• Simple and cheap</li> <li>• Rather sensitive</li> <li>• Applicable to large areas</li> <li>• Needs no special knowledge (as e.g. <b>TEM</b>)</li> </ul> | <ul style="list-style-type: none"> <li>• Black art</li> <li>• Detection limit unclear</li> <li>• What you see must be interpreted</li> <li>• Problems with artifacts</li> <li>• Mechanism not clear</li> <li>• No systematic developments of etches</li> </ul> |

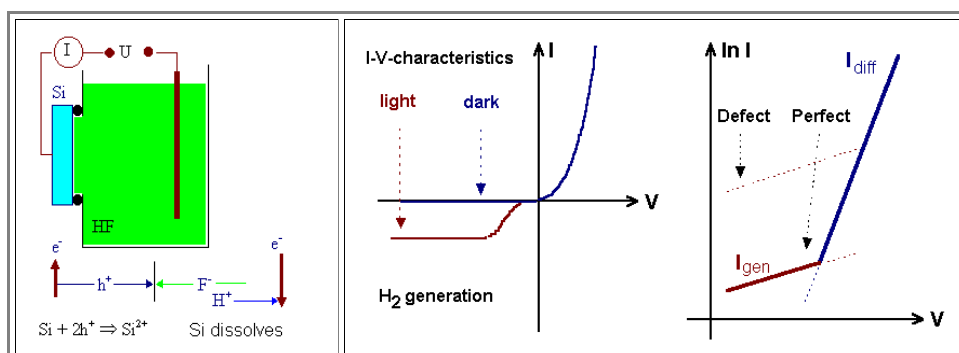
- ▶ One last example serves to illustrate the "**what you see must be interpreted**" point. Shown is a complex defect composed of stacking faults, dislocations and possibly a **microtwin** in full splendor in a **TEM** micrograph (left), and a schematic outline of what the preferential etching would look like in an optical microscope.



- The planar defects are inclined in a thin foil; what one sees is the projection. One surface was preferentially etched; at the intersection of the defect with this surface the etch features can be seen as bright areas (the sample thickness is smaller at etched parts). The stacking fault lines will be clearly visible in an etch picture, but the various dislocations involved are etched with different strengths.
- It will not be possible to conclude from the etch pattern alone on the complexity of the actual defect. This stacking fault assembly corresponds to some extent to the etch pattern shown in the development of stacking faults in bipolar patterns given [in the link](#).
- ▶ Chemical etching on occasion is driven to extremes - simply because there is no alternative. The link leads to an advanced module, where a particular [tricky case study](#) is presented

### Anodic Etching of Defects and EBIC

- ▶ Chemical etching, as any chemical dissolution process, is an oxidation-reduction process expressed in chemical terms. Carriers are transferred from the substrate to the chemicals, new compounds form and go into solution. The paradigmatical model for these processes is **anodic dissolution** under applied bias, where the carriers are supplied by a controlled external power source. Maybe a way towards the understanding of preferential etching comes from the electrochemistry of the specimen?
- ▶ Anodic etching has been studied to some extent in Silicon. It leads to a rather unexpected wealth of effects that are at the focus of some [current research projects](#). The experiment is simple:
  - Bias the (**p-type**) **Si** sample positively in some electrolyte that contains hydrofluoric acid (**HF**). The **HF** itself is "contacted" by some inert electrode, e.g. a **Pt** wire, which establishes a closed circuit.
  - The **Si-HF**- junction behaves to some extent like a **Schottky** junction; current flow, however, is always accompanied by a chemical reaction. The current density first increases steeply with the applied bias, then reaches a maximum (called  $j_{PSL}$ ; **PSL** stands for "porous **Si** layer") and decreases again (that is when the analogy with a Schottky junction fails), goes through a second maximum (called  $j_{ox}$ ) and finally starts to oscillate .
  - In the "forward" regime of the junction, the reaction is the dissolution of **Si** (in reverse condition it is **H<sub>2</sub>** evolution).
- ▶ If a polished specimen that was subjected to a current density considerably smaller than the first peak value is inspected after some etching time, its defect will be revealed in a way reminiscent of purely chemical etching. This can be understood (in parts) by considering current flow in terms of [diffusion current and generation currents](#) as introduced in basic **pn**- (or Schottky)-junction theory. The major ingredients for anodic etching are shown below.



Basic experimental set-up, current flow and chemical reaction

Measured  $I$ - $V$ -characteristic and theoretical plot of  $I_n$  / vs.  $V$  with diffusion and generation currents. Around a defect the generation current is larger than in perfect **Si**.

- ▶ Preferential defect etching thus can be understood in terms of current flow: At small current densities the generation currents are larger than the diffusion current, the area around **electronically active** defects (i.e. defects that generate carriers) should be etched more deeply and etch pits should appear. At larger current densities the differential etch rate should disappear. The experiments support this view to some extent; the link contains some results
  - [General results of anodic etching](#)
- ▶ The consideration of the influence of defects on a Schottky junction suggests a different approach to the detection of electronically active defects: Measure the local leakage current or radiation induced current of a junction. This can be done by injecting current locally by an electron beam through a thin Schottky barrier while measuring the induced current. Electronically active defects will recombine more carriers than the defect-free regions, the current will be locally reduced.
  - This method exists and is called "**electron beam induced current**" technique (**EBIC**) if a scanning electron microscope is used as the basic instrument. If a scanned light beam is used, we have the "**light beam induced current**" technique or **LBIC**; the mainstay of solar cell development with poly crystalline **Si**.
  - The [principle of EBIC](#) is shown in the link.
  - If one compares anodic etching, chemical etching and **EBIC**, much can be learned about defects and the detection methods, but many questions remain open. [Some examples](#) are given in the link
- ▶ Anodic etching is still a virulent research issue within the context of the [general electrochemistry of semiconductors](#).